

Experimental problem  
uniquely solved by TEM:  
Observation of  
dislocations and their  
dynamics

Milan Kornjaca, Physics 590B

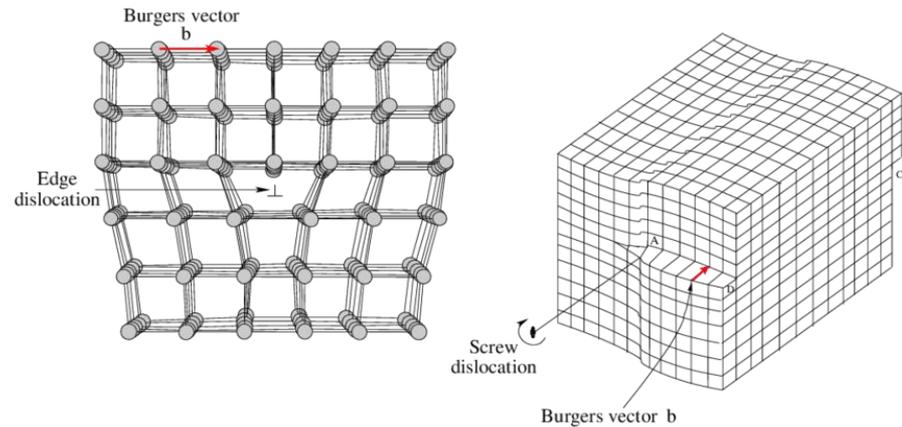
# 3000-year-old problem

- Why can metals be work-hardened and annealed?
- Why is steel harder than iron?



# Theory of dislocations

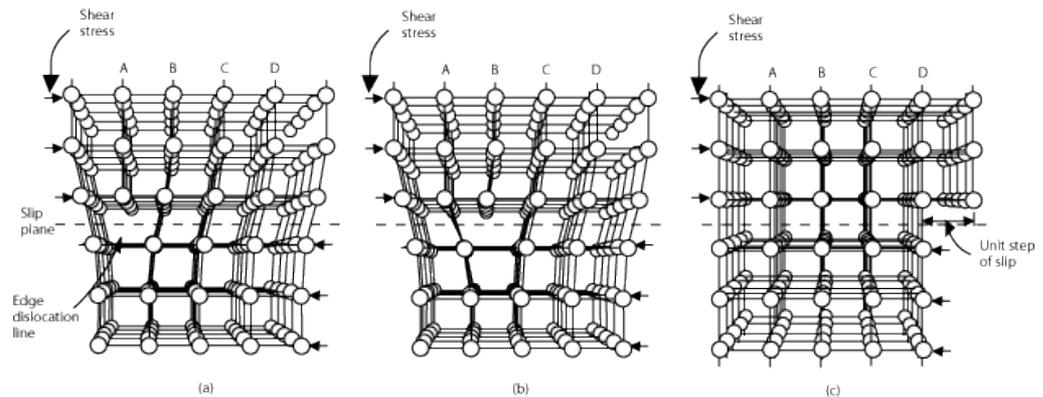
- Taylor, Orowan, and Polanyi (independently) – edge dislocation, 1934.
- Burgers – screw dislocation and Burgers vector, 1939.
- Frank and Read (independently) – formation and multiplication of dislocations (Frank-Read source), 1950.



González-Viñas, W., & Mancini, H. L. (2004). *An introduction to materials science*. Princeton, NJ: Princeton University Press.

# Theory of dislocations – movement and pinning

- perfect single crystals are very hard (compared to imperfect ones at least) – but also very rare
- dislocations are defects that can easily move through crystal and cause slippage
- they can pin on each-other (when there are many of them, like after work hardening) and other defects (like alloy precipitates), thus increasing the hardness again
- finally dislocations can be unpinned and destroyed by heating (annealing) – creep and recrystallization

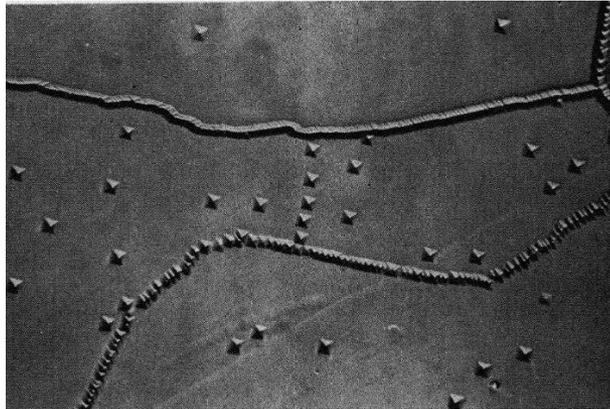


Hertzberg, R. W. (2012). *Deformation and fracture mechanics of engineering materials*. Chichester: John Wiley & Sons.

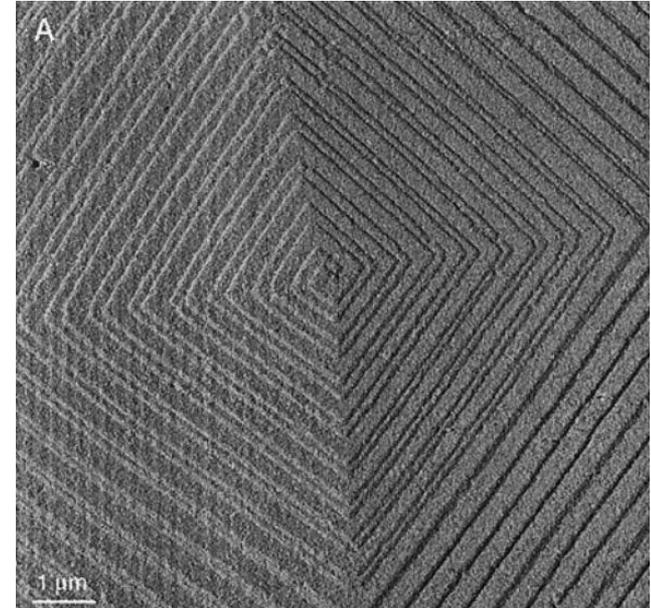
Theory was essentially complete in late 40's with little experimental evidence for existence of dislocations!

# Observation of dislocations – first attempts

- etching – observation of strain patterns at the intersection of dislocation with the surface
- observation of crystal growth around screw dislocation
- many X-ray diffraction attempts (smeared spots...), but nothing besides twinning and grain sizes



Dislocation pits on the etched surface of LiF.<sup>1</sup>



Crystal growth around a screw dislocation in calcite (AFM image).<sup>2</sup>

<sup>1</sup> J. J. Gilman and W. G. Johnston. *Dislocations and Mechanical Properties of Crystals (a report of the 1956 Lake Placid Conference)*. p. 116. John Wiley and Sons.

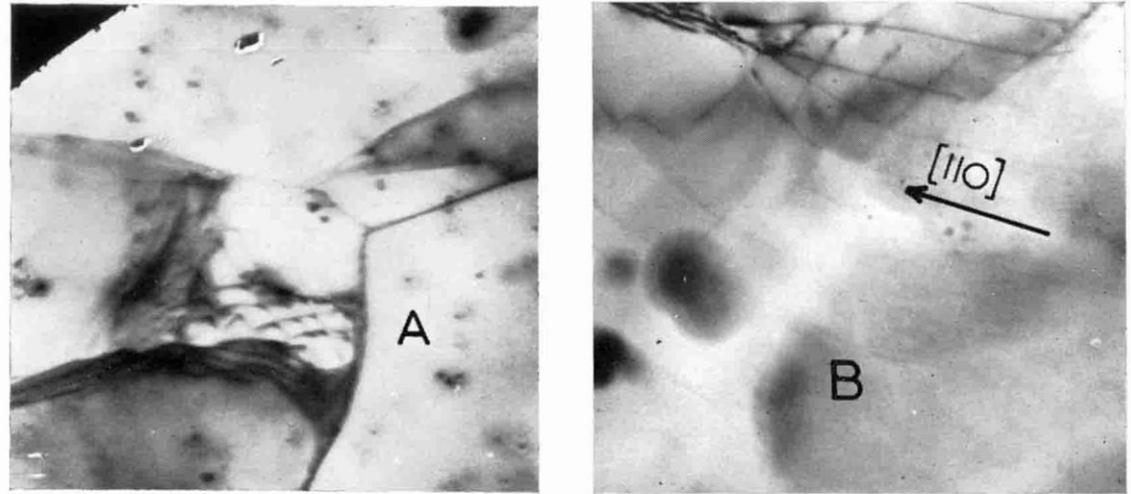
<sup>2</sup> Teng, H., Dove, P. M., Yoreo, J. J. (2000). *Geochimica Et Cosmochimica Acta*, 64(13), 2255-2266.

This was still inconclusive, at best gave only dislocations crossing the surface and the most important theoretical predictions were untestable (like moving dislocations)!

# First direct observation of dislocations - TEM

- 1955-56, Peter Hirsch, X-ray crystallographer (PhD under Bragg, the younger).
- TEM reached nanometer resolutions at the time, so considered a better prospect for finding dislocations.
- Problems – thin enough foils to get transmission (hammered gold, etched aluminium), interpretation (possibility of Moire patterns).
- First dislocations were observed in gold, but aluminium was annealed and produced nicer pictures, so it was published first!

Fig. 7



1000 Å  
└──────────┘

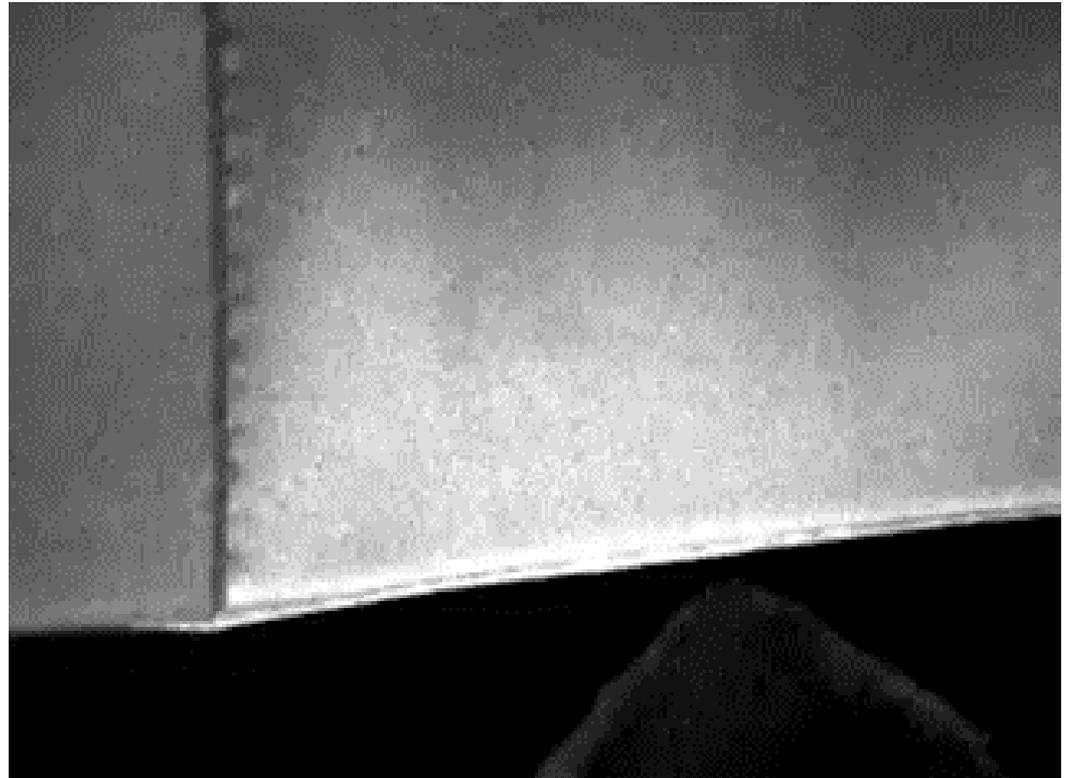
A, B:—Square cross-grids of dislocations. The dislocations in B are approximately parallel to  $[110]$  directions.

Mag.  $\times 100\ 000$

Hirsch, P. B., Horne, R. W., Whelan, M. J. (1956). LXVIII. Direct observations of the arrangement and motion of dislocations in aluminium. *Philosophical Magazine*, 1(7), 677-684.

*So the lines started running around, and from there on everything became clear.<sup>1</sup>*

- Solutions for problems
- Aluminium was beaten, annealed, then etched with HF to 0.5  $\mu\text{m}$  thickness. (Gold was just beaten.)
- Tilting the sample changed the contrast, but lines were still there.
- Switching to higher brightness setting (not suitable for diffraction) revealed moving lines!
- Moving dislocations showed many of the characteristics predicted by theory (movement along glide planes, Frank-Read sources).



2

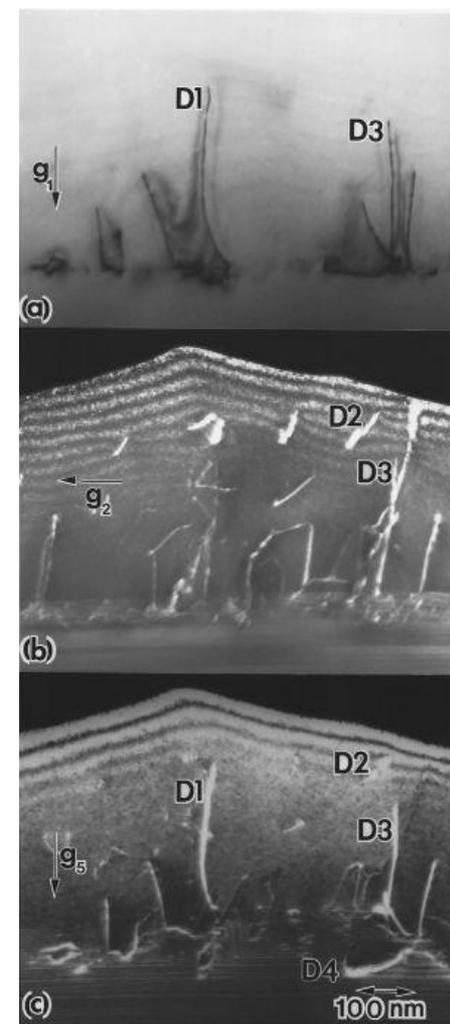
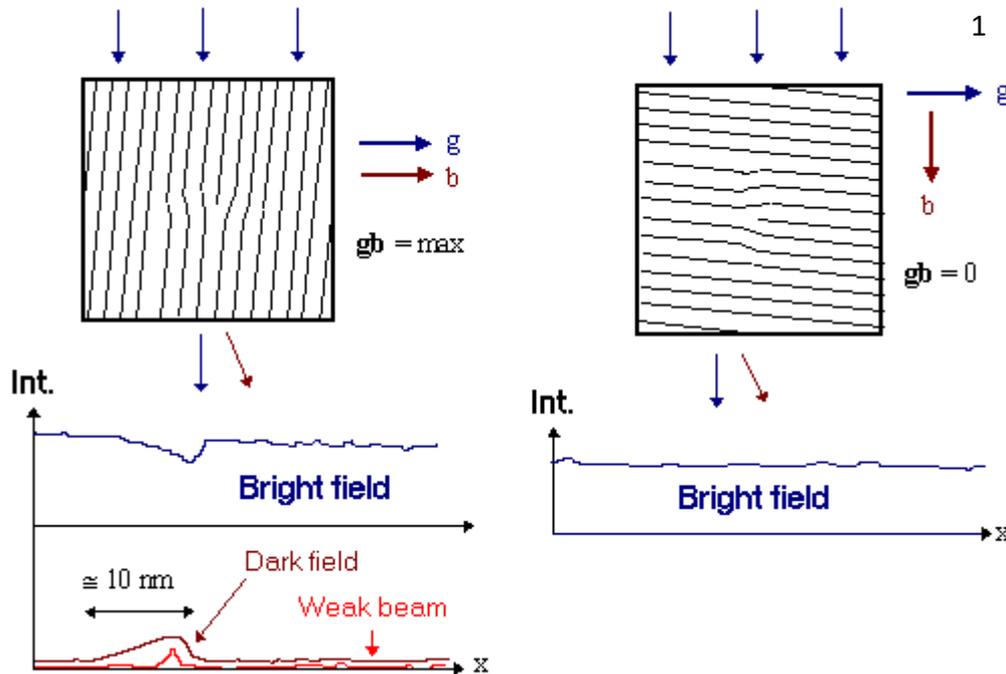
Generation, movement and pinning of dislocations near grain boundary in  $\text{SrTiO}_3$ .

<sup>1</sup> AIP Oral History Interviews: Interview with Peter Hirsch.

<sup>2</sup> Kondo, S., Mitsuma, T., Shibata, N., & Ikuhara, Y. (2016). Direct observation of individual dislocation interaction processes with grain boundaries. *Science Advances*, 2(11).

# TEM – still by far the leading technique for dislocation imaging

- Quantitative results are possible as well – like Burgers vector analysis.
- Intensity of beam diffracted from dislocation is proportional to  $\vec{g} \cdot \vec{b}$ .

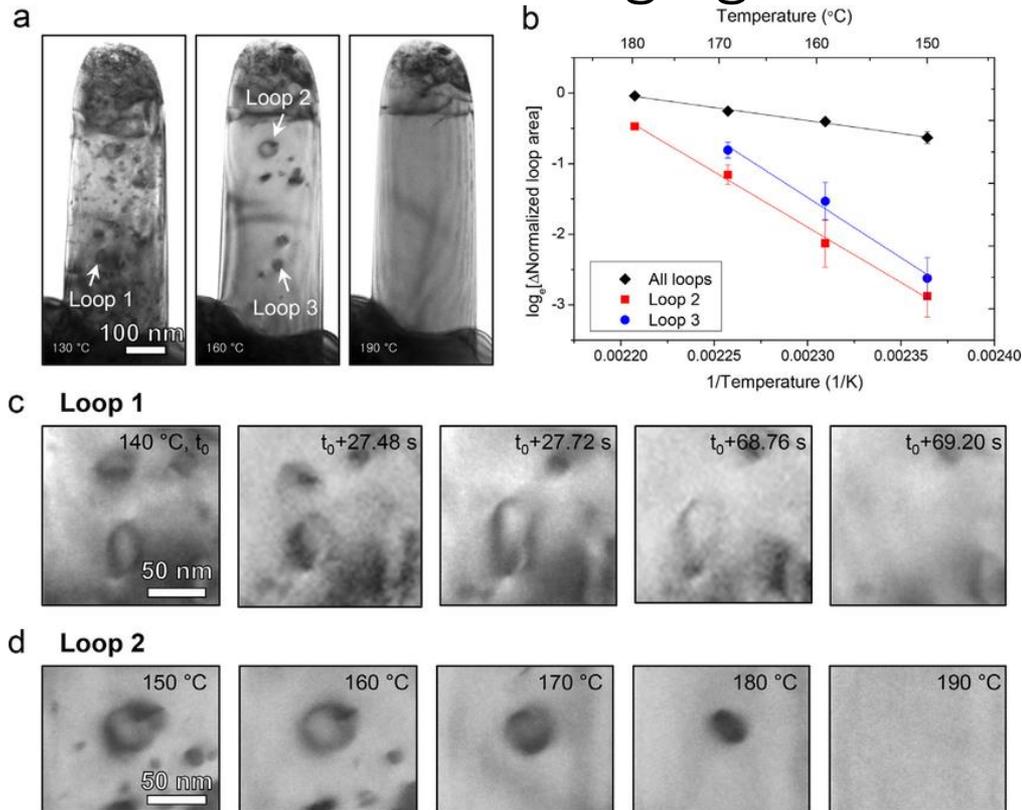


Burgers vector analysis of dislocations in GaN film. (a) is taken in bright field mode, while (b) and (c) are taken in dark field mode with different diffraction vectors.<sup>2</sup>

<sup>1</sup> [https://www.tf.uni-kiel.de/matwis/amat/def\\_en/index.html](https://www.tf.uni-kiel.de/matwis/amat/def_en/index.html)

<sup>2</sup> Cherns, D., Young, W., & Ponce, F. (1997). *Materials Science and Engineering: B*, 50(1-3), 76-81.

# TEM – still by far the leading technique for dislocation imaging



Annihilation of dislocations by annealing in a pillar of Al. Case (c) shows a dislocation that vanished by “escaping through the surface”. Case (d) shows a dislocation loop that shrank by diffusion of vacancies. Rate of heating is 5K/min.

Thank you!